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Molecular Engineering of Liquid Crystal Polymers by Living Polymerization.5.  
Synthesis and mesomorphic Behavior of Poly(2-[4-Cyano-4'-biphenyl]oxy)ethyl Vinyl  
Ether-co-8[4-Cyano-4'-biphenyl]oxy]octyl Vinyl Ether}

by

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**Molecular Engineering of Liquid Crystal Polymers by Living Polymerization 5<sup>a</sup>.**

**Synthesis and Mesomorphic Behavior of Poly{2-[(4-cyano-4'-biphenyl)oxy]ethyl  
vinyl ether-co-8-[(4-cyano-4'-biphenyl)oxy]octyl vinyl ether}**

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## ABSTRACT

The synthesis and characterization of poly{2-[(4-cyano-4'-biphenyl)oxy]ethyl vinyl ether} [poly(6-2)], poly{8-[(4-cyano-4'-biphenyl)oxy]octyl vinyl ether}[poly(6-8)] and of poly (6-2-co-6-8)X/Y (where X/Y is the molar ratio of the two monomeric structural units) with degrees of polymerization of about 10 and narrow molecular weight distribution are described. Both homopolymers and copolymers were prepared by the living cationic polymerization and copolymerization of 6-2 and 6-8. During the first heating scan poly(6-2) presents an inverse monotropic  $s_C$  mesophase. During subsequent heating and cooling scans, this polymer is amorphous regardless of the thermal history of the sample. Poly(6-8) exhibits an enantiotropic  $s_A$  mesophase. Both during the first and subsequent heating and cooling scans, the phase diagram which plots the dependence of mesomorphic transition temperatures as a function of copolymer composition presents a discontinuity. During the first heating scan, poly(6-2-co-6-8)X/Y with X/Y=10/0 to 6/4 represents an inverse  $s_C$  monotropic mesophase. Poly(6-2-co-6-8)X/Y with X/Y=8/2 to 6/4 show in addition to the  $s_C$  phase an enantiotropic nematic mesophase. Poly(6-2-co-6-8) 5/5 exhibits only an enantiotropic nematic mesophase, while poly(6-2-co-6-8)4/6 displays both a nematic and a  $s_A$  enantiotropic mesophase. Poly(6-2-co-6-8)X/Y with X/Y=3/7 to 0/10 present an enantiotropic  $s_A$  mesophase. The second and subsequent heating scans and first and subsequent cooling scans provide an identical phase behavior with that obtained from the first heating scan, except that the  $s_C$  mesophase is absent.

(Keywords: side chain liquid crystalline polyethers and copolyethers, living cationic polymerization)

## INTRODUCTION

Copolymerization represents the simplest synthetic technique which can be used to tailor make phase transitions of both main chain and side chain liquid crystalline polymers.<sup>1</sup> Indeed, copolymerization was frequently employed to manipulate the phase transitions of side chain liquid crystalline polymers. However, most of the results reported in the literature can be considered only in a qualitative way, since there are only very few cases in which information about both copolymer composition and molecular weight are available.<sup>1-6</sup>

A general classification of side chain liquid crystalline copolymer was recently discussed in a review article.<sup>1</sup> There are two main classes of side chain liquid crystalline copolymers. The first one refers to copolymers containing pairs of structural units with and without mesogenic units.<sup>1-4,6-12</sup> This class of copolymers was extensively investigated. The second class refers to copolymers based on pairs of structural units containing mesogenic units in each one. These are at least four different categories which should be considered in this second class; a) copolymers from monomer pairs containing identical mesogens and polymerizable groups, but different spacer lengths; b) copolymers from monomer pairs containing identical mesogens and spacer lengths but different polymerizable groups; c) copolymers from monomer pairs containing dissimilar mesogens, but either similar or different spacer lengths and polymerizable groups; and d) copolymers from monomer pairs containing constitutional isomeric mesogenic units, and similar or dissimilar spacers and polymerizable groups. Presently, there is a relatively good understanding of the last class of copolymers.<sup>13,14</sup> When the structural units of these copolymers are isomorphic in their liquid crystalline phase but not in their crystalline phase, copolymerization could be used to transform virtual or monotropic mesophases into enantiotropic mesophases. A general discussion considering the isomorphism in liquid crystalline polymers and copolymers is provided in reference.<sup>15</sup>

The following general trends were observed so far when copolymers from the second class of type a,b and c were investigated. When the structural units of the copolymer were isomorphic within the liquid crystalline phase, a continuous or even linear dependence of the phase transition temperature versus copolymer composition was observed.<sup>16</sup> When the structural units of the copolymer were nonisomorphic within the mesophase, a discontinuous dependence of phase transitions versus composition was observed.<sup>16-19</sup>

Although the composition of these copolymers were reported, there was no information on the molecular weight of any of them. Liquid crystalline copolysiloxanes

are prepared by polymer homologues reactions. They are considered to be statistical copolymers with a random distribution of their structural units.<sup>16</sup> Copolymers synthesized by classic polymerization reactions exhibit a heterogeneous composition unless they are prepared at low conversions.<sup>16</sup> At the same time, there is no single example in the literature in which the molecular weights of chain copolymers were reported.<sup>1,3,5,17-19</sup> Since mesomorphic transitions are molecular weight dependent,<sup>1,11,20-28</sup> a quantitative understanding of our ability to tailor make liquid crystals by copolymerization requires the synthesis of copolymers with both well defined compositions and molecular weights. The ideal solution to the synthesis of copolymers by chain reactions would be to select monomer pairs which follow an azeotropic copolymerization and can be prepared by a living polymerization mechanism. Such systems require  $r_1=r_2=1$  and are encountered mainly for comonomer pairs of almost similar structure.

Vinyl ethers containing an identical mesogenic group but different spacer lengths are presently the most suitable monomers for these investigations.<sup>23-29</sup> The living cationic polymerization of 2-[(4-cyano-4'-biphenyl)oxy]ethyl vinyl ether,<sup>25</sup> 11-[(4-cyano-4'-biphenyl)oxy]undecanyl vinyl ether,<sup>26</sup> 8-[(4-cyano-4'-biphenyl)oxy]octyl vinyl ether<sup>27</sup> and of 6-[(4-cyano-4'-biphenyl)oxy]hexyl vinyl ether<sup>28</sup> as well as the characterization of the resulting polymers as a function of molecular weight were already investigated.

The first goal of this paper is to present the synthesis of the copolymers of 2-[(4-cyano-4'-biphenyl)oxy]ethyl vinyl ether and 8-[(4-cyano-4'-biphenyl)oxy]octyl vinyl ether with different compositions and a degree of polymerization of 10. The second goal is to describe the phase behavior of these copolymers as a function of copolymer composition. Poly{2-[(4-cyano-4'-biphenyl)oxy]ethyl vinyl ether} with a degree of polymerization equal to 10 displays a  $s_C$  mesophase only during the first heating scan.<sup>26</sup> Cooling and subsequent heating scans provide only an amorphous polymer.<sup>25</sup> Poly{8-[(4-cyano-4'-biphenyl)oxy]octyl vinyl ether} with a degree of polymerization equal to 10 displays an enantiotropic  $s_A$  phase, regardless of the thermal history of the sample.<sup>27</sup> Therefore, it is expected that the investigation of this copolymer, during the first heating scan, will provide information about the isomorphism of the two structural units whose homopolymers display  $s_C$  and  $s_A$  mesophases respectively.

Alternatively, the investigation of this copolymer, during the cooling and second and subsequent heating scans, will provide information about the isomorphism of the two structural units whose parent homopolymers display a glassy and a  $s_A$  phase

respectively. Both structural units of these copolymers contain the same mesogenic group but different spacer lengths.

## EXPERIMENTAL

### Materials

4-Phenylphenol (98%), 1,10-phenanthroline (anhydrous, 99%) palladium (II) acetate (all from Lancaster Synthesis), ferric chloride anhydrous (98%), copper(I)cyanide (99%), n-butyl vinyl ether (98%), 9-borabicyclo[3.3.1]nonane (9-BBN, crystalline, 98%), 8-bromo-1-octanol (95%) (all from Aldrich), 2-chloro ethyl vinyl ether (Polysciences, bp, 109-110°C) and the other reagents were used as received. Methyl sulfide (anhydrous, 99%, Aldrich) was refluxed over 9-BBN and then distilled under argon. Dichloromethane (99.6%, Aldrich) used as polymerization solvent was first washed with concentrated sulfuric acid, then with water, dried over anhydrous magnesium sulfate, refluxed over calcium hydride and freshly distilled under argon before each use. N-Methyl-2-pyrrolidone (98%, Lancaster Synthesis) was dried by azeotropic distillation with benzene, shaken with barium oxide, filtered, and fractionally distilled under reduced pressure. Trifluoromethane sulfonic acid (triflic acid, 98%, Aldrich) was distilled under argon.

### Techniques

<sup>1</sup> H-NMR (200 MHz) spectra were recorded on a Varian XL-200 spectrometer. TMS was used as internal standard. A Perkin Elmer DSC-4 differential scanning calorimeter, equipped with a TADS 3600 data station, was used to determine the thermal transitions which were reported as the maxima and minima of their endothermic and exothermic peaks respectively. In all cases, heating and cooling rates were 20°C/min unless otherwise specified. Glass transition temperatures (T<sub>g</sub>) were read at the middle of the change in the heat capacity. For certain samples, first heating scans differ from second and subsequent heating scans. However, second and subsequent heating scans are identical. The first heating scans can be reobtained after proper thermal treatment of the polymer sample. A Carl-Zeiss optical polarized microscope (magnification 100X) equipped with a Mettler FP 82 hot stage and a Mettler FP 800 Central Processor was used to observe the thermal transitions and to analyze the anisotropic textures.<sup>30,31</sup> Molecular weights were determined by gel permeation chromatography (GPC) with a Perkin-Elmer Series 10 LC instrument equipped with LC-100 column oven, LC-600 autosampler and a Nelson Analytical 900 series intergrator data station. The

measurements were made at 40°C using the UV detector. A set of Perkin-Elmer PL gel columns of 10<sup>4</sup> and 500Å with chloroform as solvent (1ml/min) and a calibration plot constructed with polystyrene standards was used to determine the molecular weights. High pressure chromatography experiments were performed with the same instrument.

### Synthesis of monomers

Scheme I outlines the synthesis of 2-[(4-cyano-4'-biphenyl)oxy]ethyl vinyl ether (**6-2**) and of 8-[(4-cyano-4'-biphenyl)oxy]octyl vinyl ether (**6-8**). 1,10-Phenanthroline palladium (II) diacetate (**9**) and 4-cyano-4'-hydroxybiphenyl (**5**) were synthesized as described previously.<sup>26-28</sup>

#### 2-[(4-Cyano-4'-biphenyl)oxy]ethyl vinyl ether (**6-2**)

A mixture of 2g (10.25mmol) 4-cyano-4'-hydroxybiphenyl, 3.16ml (32.8mmol) 2-chloroethyl vinyl ether, 0.45g (11.07mmol) of powdered NaOH, 0.176g of tetrabutylammonium hydrogen sulfate (0.533mmol), 10.5ml toluene and 2ml dimethyl sulfoxide was stirred at 90°C for 6 hr. After cooling, the reaction mixture was washed with dilute NaOH, water, and dried over MgSO<sub>4</sub>. The solvent was evaporated in a rotavapor and the resulting solid was recrystallized from methanol to yield 2.3g (87%) of white crystals which were further purified by column chromatography (silica gel, CH<sub>2</sub>Cl<sub>2</sub> eluent). Purity: 99% (HPLC). mp, 104-105°C (lit 25, mp, 105°C). <sup>1</sup>H-NMR (CDCl<sub>3</sub>, TMS, δ, ppm): 4.07 (3 protons, -CH<sub>2</sub>OCH=CH<sub>2</sub> trans, m), 4.27 (3 protons, PhOCH<sub>2</sub> and -OCH=CH<sub>2</sub> cis, m), 6.55 (1proton, -OCH=CH<sub>2</sub>, q), 7.02 (2 aromatic protons, o to alkoxy, d), 7.53 (2 aromatic protons, m to alkoxy, d), 7.69 (4 aromatic protons, o and m to -CN, d of d).

#### 8-[(4-Cyano-4'-biphenyl)oxy]octyl vinyl ether (**6-8**)

**6-8** was synthesized by the transesterification of 4-cyano-4'-(8-hydroxyoctan-1-yloxy)biphenyl (**7-8**) with n-butyl vinyl ether by using 1,10-phenanthroline palladium (II) diacetate as catalyst. Its detailed synthesis was presented in a previous publication.<sup>27</sup> Purity: 99.9% (HPLC). T<sub>k-n</sub>=54.0°C, T<sub>n-i</sub>=70.8°C.

### Cationic Polymerizations

Polymerizations and copolymerizations were carried out in glass flasks equipped with teflon stopcocks and rubber septa under argon atmosphere at 0°C for 1 hr. All glassware was dried overnight at 130°C. The monomer(s) was further dried under vacuum overnight in the polymerization flask. Then the flask was filled with argon.

cooled to 0°C and the required amounts of methylene chloride and triflic acid were added via a syringe. The monomer concentration was about 10 wt% of the solvent volume and the dimethyl sulfide concentration was 10 times larger than that of the initiator. The molecular weight was controlled by the monomer/initiator ratio. At the end of the polymerization, the reaction mixture was precipitated in methanol containing few drops of NH<sub>4</sub>OH. The filtered polymers were dried and precipitated from methylene chloride solutions into methanol until GPC traces showed no trace of monomer. Table I summarizes the polymerization and copolymerization results. Although the polymer yields are lower than expected due to losses during the purification process, the conversions were almost quantitative in all cases.

## RESULTS AND DISCUSSION

Scheme II presents the living cationic copolymerization of 2-[(4-cyano-4'-biphenyl)oxy]ethyl vinyl ether (6-2) with 8-[(4-cyano-4'-biphenyl)oxy]octyl vinyl ether (6-8). An extensive study on the living cationic polymerization of 6-8 and on the characterization of the mesomorphic phase behavior of poly(6-8) as a function of molecular weight was presented in a previous publication.<sup>27</sup> All copolymerization experiments described in this paper were performed under experimental conditions where the resulting poly(6-2), poly(6-8) and poly(6-2-co-6-8) have a degree of polymerization of about 9-10 (Table I). The yields reported in Table I are lower than 80% due to polymer losses during purification process. However, the conversions were quantitative and therefore, the copolymer composition is identical to that of the monomer feed. 200 MHz <sup>1</sup>H-NMR spectra of the copolymer samples support this statement. Poly (6-8) with degree of polymerization equal to 10 displays an enantiotropic SA mesophase<sup>27</sup> (Table I, Figure 1). Poly (6-2) was previously synthesized and characterized by Sagane and Lenz.<sup>25</sup> Poly(6-2) with a degree of polymerization of 10 displays only in the first heating scan an inverse monotropic SC phase.<sup>25</sup> During the first and subsequent cooling scans and second and subsequent heating scans, poly (6-2) displays only a glass transition temperature (Table I, Figure 1). Our results agree with these reported by Sagane and Lenz.<sup>25</sup> Therefore, the investigation of copolymers based on the monomer pair 6-2 and 6-8 provides a unique system which can give information about the phase behavior of a copolymer based on a structural unit whose homopolymer exhibits a SC mesophase and a structural unit whose homopolymer exhibits a SA mesophase when the data are collected from the first heating scan. Alternatively, when

the data are collected from the second heating scan, the same copolymer system consists of a structural unit whose homopolymer is amorphous and a structural unit whose homopolymer exhibits an enantiotropic  $s_A$  mesophase. Consequently, poly(6-2-co-6-8) represents a unique binary copolymer which from a single series of experiments provides information about the phase behavior of two "different" binary copolymers.

Let us first discuss the behavior of this copolymer as being determined from the first DSC heating scans. The first DSC heating scans of poly(6-2-co-6-8) are presented in Figure 1a. The nature of the mesophase displayed by various copolymers is presented on the figure. Poly(6-2) displays a  $s_C$  phase which overlaps the glass transition temperature (Table I). Upon copolymerization, the temperature associated with the  $s_C$ -isotropic phase transition decreases. Copolymers poly(6-2-co-6-8)5/5 to poly(6-2-co-6-8)0/10 do not exhibit the  $s_C$  phase. Therefore, the structural units derived from the monomers 6-2 and 6-8 are isomorphic within the  $s_C$  phase only over a very narrow range of compositions. A similar behavior can be observed for the  $s_A$  phase displayed by poly(6-8). Poly(6-2-co-6-8)X/Y with compositions from X/Y=0/10 to 4/6 display a  $s_A$  phase. The temperature transition associated with this phase decreases by increasing the concentration of structural units derived from monomer 6-2. Poly(6-2-co-6-8)4/6 exhibits both a nematic and a  $s_A$  mesophase. Poly(6-2-co-6-8)X/Y with X/Y from 5/5 to 8/2 exhibits a nematic phase.

The phase behavior of poly(6-2-co-6-8)X/Y determined from the first heating scan is plotted in Figure 2a. This figure demonstrates that both the  $s_A$  and  $s_C$  phases exhibit discontinuous dependences of copolymer composition. The very interesting result consist in the ability to prepare copolymers exhibiting a nematic mesophase from structural units derived from two homopolymers which both display a smectic mesophase. This may be the result of the fact that within the  $s_A$  phase the backbone conformation of poly(6-8) is confined to the smectic layers.<sup>12</sup> Simultaneously, since the spacer of poly(6-2) is short, the backbone of this polymer may exhibit an extended conformation within the  $s_C$  phase.<sup>1</sup> Based on this account, the structural units of this copolymer are isomorphic in each of the two mesophases only over the range of compositions where the polymer backbone can get distorted to accomodate either the  $s_C$  or  $s_A$  mesophases. Morphological experiments are required to support this speculative explanation. The newly formed nematic mesophase extends over the broadest range of copolymer composition. The glass transition temperature of these copolymers displays an expected continuous dependence of copolymer composition (Figure 2a).

The DSC traces of the second heating scans of poly(6-2-co-6-8)X/Y are presented in Figure 1b. They are identical to those from the first heating scans from Figure 1a.

except that the SC phase is missing and the change in the  $\Delta C_p$  at the glass transition temperatures of poly(6-2-co-6-8)X/Y from X/Y= 10/0 to 6/4 is larger. The thermal transition temperatures collected from Figure 1b are plotted in Figure 2b. This Figure demonstrates that poly(6-2-co-6-8) displays a discontinuous dependence of the SA and nematic phases on copolymer composition. The absence of the SC phase in the second heating scans is most probable the result of a kinetic effect which is due to the close proximity of the SC phase from the glass transition. Therefore this SC phase forms only when the polymers are precipitated from solution. If the polymer sample is redissolved and reprecipitated, the SC phase reappears. Alternatively if the sample of poly(6-2) and poly(6-2-co-6-8)9/1 is sheared above the glass transition temperature on the optical polarized microscope, an anisotropic texture forms. This behavior can be easily explained by thermodynamics.<sup>32,33</sup>

The cooling DSC scans of poly(6-2-co-6-8)X/Y are presented in Figure 1c. They provide the same conclusion as that derived from the second heating scans. The phase diagram of poly(6-2-co-6-8) obtained from the cooling DSC scans is plotted in Figure 2c and agrees very well with that from Figure 2b.

The dependence of the enthalpy changes associated with nematic-isotropic, isotropic-nematic, SA-isotropic and isotropic-SA is plotted in Figure 3. The enthalpy changes of the nematic-SA and SA-isotropic as well as their reversed values of poly(6-2-co-6-8)4/6 could not be separated and therefore, they were plotted as the sum of both transitions (Table I, Figure 3). Even so, Figure 3 demonstrates that both the enthalpy changes associated with the SA-phase and with the nematic phase transitions are located on straight lines. However, these lines have a different slope. These data prove that the structural units derived from 6-2 and 6-8 are isomorphic over a certain range of copolymer composition within the nematic phase and over different range of composition within the SA-phase. A classification of the various classes of homopolymer and copolymer isomorphism within liquid crystalline phases was presented in a previous publication.<sup>34</sup> The plot from Figure 3 shows that the mesomorphic phase behavior of poly(6-2-co-6-8)X/Y versus copolymer composition is not continuous. An attempt to extrapolate the linear dependence of the enthalpy change associated with the nematic-isotropic and isotropic-nematic phase transition to poly(6-2) leads to a value of  $\Delta H=0$ . This result demonstrates that poly(6-2) with a degree of polymerization of 10 does not exhibit a nematic mesophase. The same conclusion can be derived if we try to extrapolate the dependences of the temperature transition of nematic-isotropic (Figure 1a,b) and isotropic-nematic (Figure 1c) versus copolymer composition to the composition corresponding to poly(6-2). We would obtain a virtual nematic mesophase which is

located below the glass transition temperature of poly(6-2). However, since the enthalpy change of this virtual mesophase as determined from Figure 3 is equal to zero, we can safely conclude that poly(6-2) with a degree of polymerization of 10 does not display a virtual nematic mesophase.

Finally, Figure 4 presents some representative textures characteristic for the SA and nematic phases exhibited by poly(6-2-co-6-8)X/Y copolymers and by poly(6-8). No characteristic texture could be obtained for the SC phase of poly(6-2) because this phase overlaps the glass transition temperature of the polymer and therefore the polymer is very viscous within its SC phase.

In conclusion, the experimental results described in this paper provide the first series of quantitative copolymerization experiments performed with mesogenic vinyl ethers. Since these monomers can be polymerized by living cationic polymerization, both the molecular weight, molecular weight distribution and the composition of these copolymers can be conveniently controlled. The most important conclusions obtained from the copolymerization of 6-2 with 6-8 are as follows. Although the two structural units of these copolymers lead to homopolymers which, depending on the thermal history of the sample, exhibit either a monotropic SC or a glassy phase and an enantiotropic SA phase, respectively, the resulting copolymers display a nematic mesophase over a quite broad range of copolymer compositions. In addition, the dependence of mesomorphic phase transitions on copolymer composition is discontinuous. The generation of a nematic polymer by copolymerization of two monomers containing dissimilar spacer lengths provides a simple technique for the synthesis of nematic polymers containing 4-cyanobiphenyl mesogenic groups. Very few examples of side chain liquid crystalline polymers containing a cyano substituent and displaying a nematic phase are available.<sup>1,35</sup> Most frequently, nematic polymers containing cyano groups are synthesized by inserting a lateral substituent into the structure of the mesogenic unit,<sup>36</sup> by using laterally attached mesogenic units,<sup>37</sup> and by copolymerization of mesogenic units containing laterally and terminally attached mesogenic side groups.<sup>38</sup>

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FIGURE AND SCHEME CAPTIONS

**Scheme 1:** Synthesis of 2-[(4-cyano-4'-biphenyl)oxy]ethyl vinyl ether (6-2) and of 8-[(4-cyano-4'-biphenyl)oxy]octyl vinyl ether (6-8).

**Scheme 2:** Cationic copolymerization of 6-2 with 6-8.

**Figure 1:** Differential Scanning Calorimetric traces displayed during the first heating scan (a), second heating scan (b), and first cooling scan (c) of poly(6-2), poly(6-8) and of poly(6-2-co-6-8) X/Y.

**Figure 2:** The dependence of phase transition temperatures on the composition of poly(6-2-co-6-8) copolymers. a) data from first heating scan: O-T<sub>g</sub>; ●-T<sub>SC</sub>-i(n); □-T<sub>n</sub>-i; Δ-T<sub>SA</sub>-n(i); b) data from second heating scan: ●-T<sub>g</sub>; ■-T<sub>n</sub>-i; Δ-T<sub>SA</sub>-n(i), c) data from first cooling scan: ■-T<sub>g</sub>; ●-T<sub>i</sub>-n; ▲-T<sub>n</sub>(i)-s<sub>A</sub>.

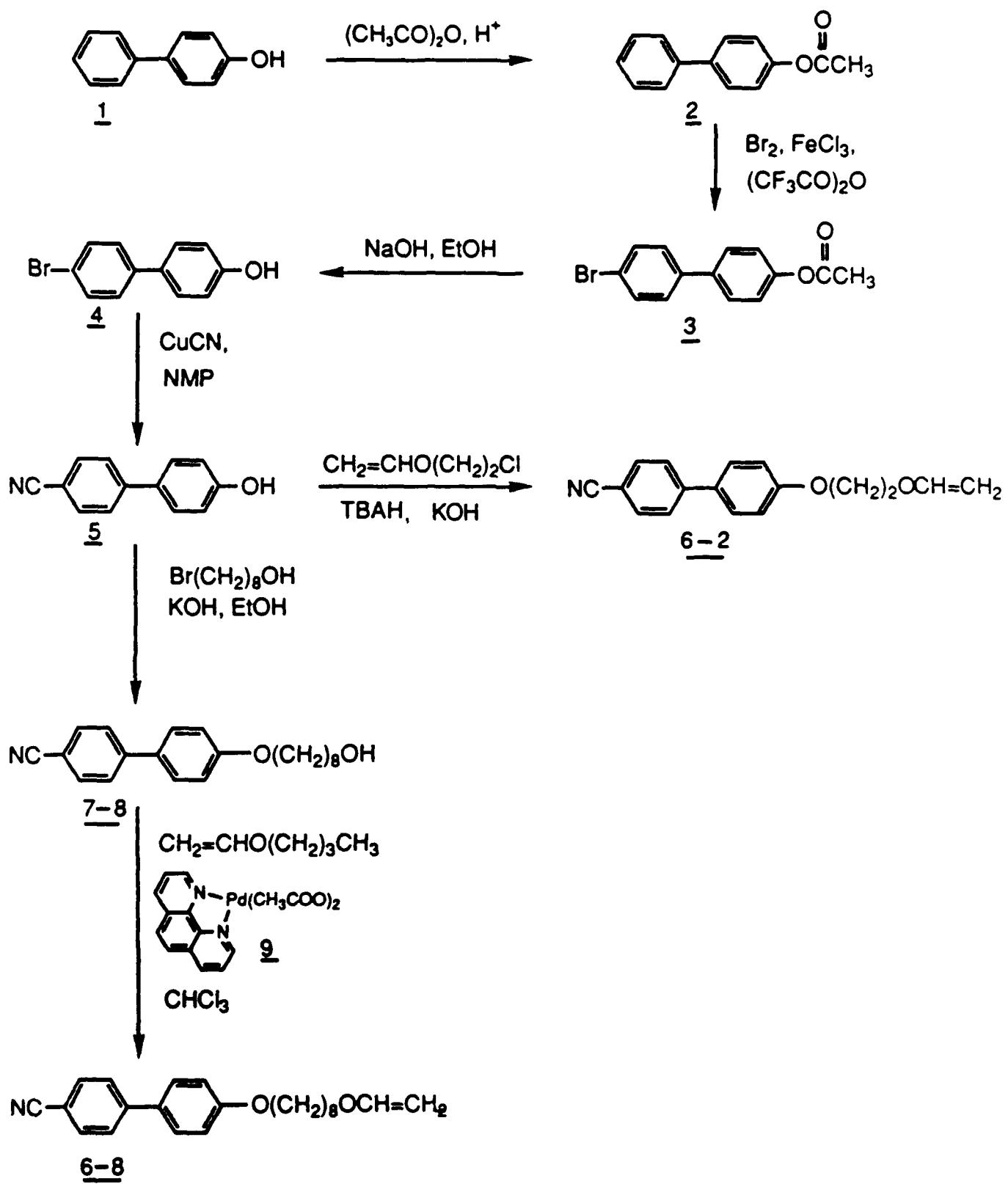
**Figure 3:** The dependence of the enthalpy changes associated with the mesomorphic-isotropic and isotropic-mesomorphic phase transitions on the composition of poly(6-2-co-6-8). □-ΔH<sub>n</sub>(s<sub>A</sub>)-i (data from the first heating scan); O-ΔH<sub>n</sub>(s<sub>A</sub>)-i (data from second heating scan); Δ-ΔH<sub>i</sub>-n and ▲-ΔH<sub>i</sub>-s<sub>A</sub> (data from first cooling scan).

**Figure 4:** Representative optical polarized micrographs (100X) of: a) the nematic phase displayed by poly(6-2-co-6-8)4/6 at 100°C on the cooling scan; b) the s<sub>A</sub> mesophase displayed by poly(6-2-co-6-8)4/6 at 90°C on the cooling scan; c) the nematic mesophase displayed by poly(6-2-co-6-8)1/1 at 90°C on the cooling scan.

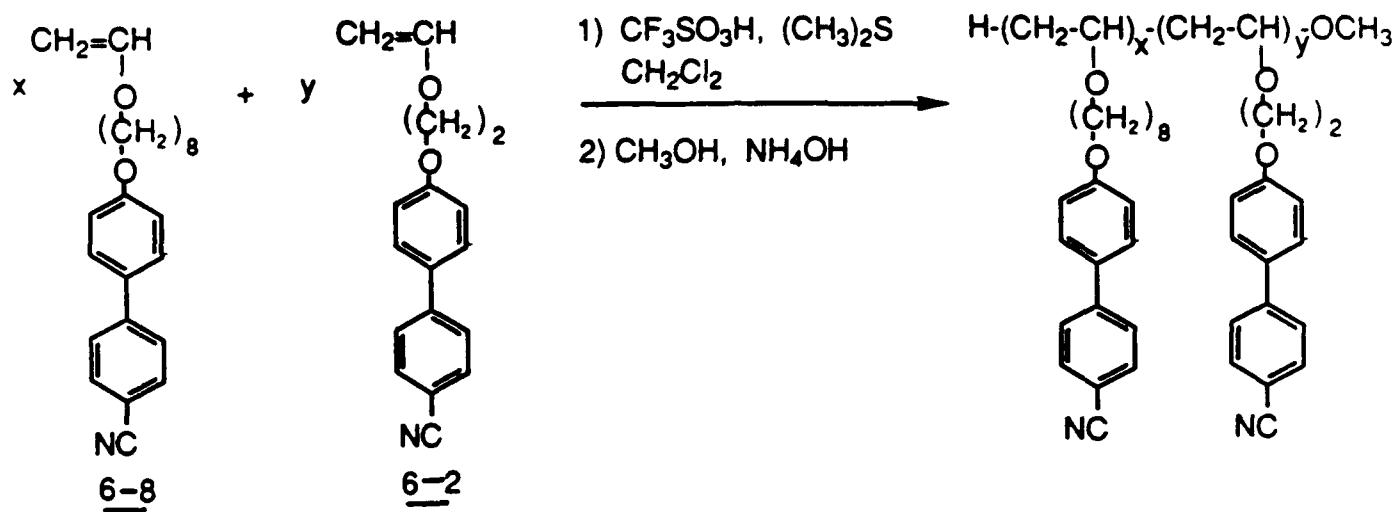
**Table I. Cationic Polymerization of 6-2 with 6-8 (polymerization temperature, 0°C; polymerization solvent,  $\text{CH}_2\text{Cl}_2$ ;  $[\text{M}]_0 = [\text{6-2} + \text{6-8}] = 0.285 - 0.377 \text{M}$ ;  $[\text{M}]_0/[\text{I}]_0 = 10$ ;  $[(\text{CH}_3)_2\text{S}]_0/[\text{I}]_0 = 10$ ; polymerization time, 1 hr) and Characterization of the Resulting Copolymers.**

Sample no.	$[\text{6-8}]/[\text{6-2}]$ (mol/mol)	Polymer yield (%)	GPC	phase transitions ( $^{\circ}\text{C}$ ) and corresponding enthalpy changes (kcal/mru)							
				$\text{Mn} \times 10^{-3}$	$\text{Mw/Mn}$	DP	heating	cooling			
1	0/10	73.1	2.87	1.10	10.8	86.0 (0.18)	i	63.8	g		
2	1/9	72.7	2.81	1.16	10.3	8C 66.3 (0.38)	i	49.8	g		
3	2/8	71.5	2.98	1.10	10.6	8C 60.8 (0.41)	n 75.2 (0.045)	i	70.8 (0.042)	n 41.0	g
4	3/7	81.3	2.87	1.13	9.8	8C 54.6 (0.46)	n 82.7 (0.049)	i	80.2 (0.053)	n 30.0	g
5	4/6	64.2	2.79	1.15	9.3	8C 45.2 (0.39)	n 94.1 (0.096)	i	90.3 (0.087)	n 27.8	g
6	5/5	65.1	2.82	1.17	9.2	9 36.0 n 93.8 (0.092)	i	98.7 (0.110)	n 22.6	g	
7	6/4	71.8	3.09	1.15	9.3	9 31.5 n 102.5 (0.110)	i	100.7 (0.18)	n 90.8 (·)	sA 13.7	g
8	7/3	75.4	3.33	1.13	10.3	9 24.9 sA 98.1 (·)	*106.5 (0.16)*i	i	104.8 (0.27)	sA 12.5	g
9	8/2	70.0	3.26	1.10	9.8	9 21.3 sA 94.6 (·)	*104.2 (0.21)*i	i	110.5 (0.35)	sA 10.1	g
10	9/1	71.1	3.66	1.10	10.7	9 18.2 sA 109.1 (0.26)	i	115.2 (0.39)	sA 10.8	g	
11	10/0	82.3	3.84	1.15	10.4	9 16.3 sA 117.0 (0.34)	i	122.8 (0.46)	sA 7.8	g	
						g 11.6 sA 127.3 (0.45)	i				

\* overlapped peaks



Scheme 1



Scheme 2

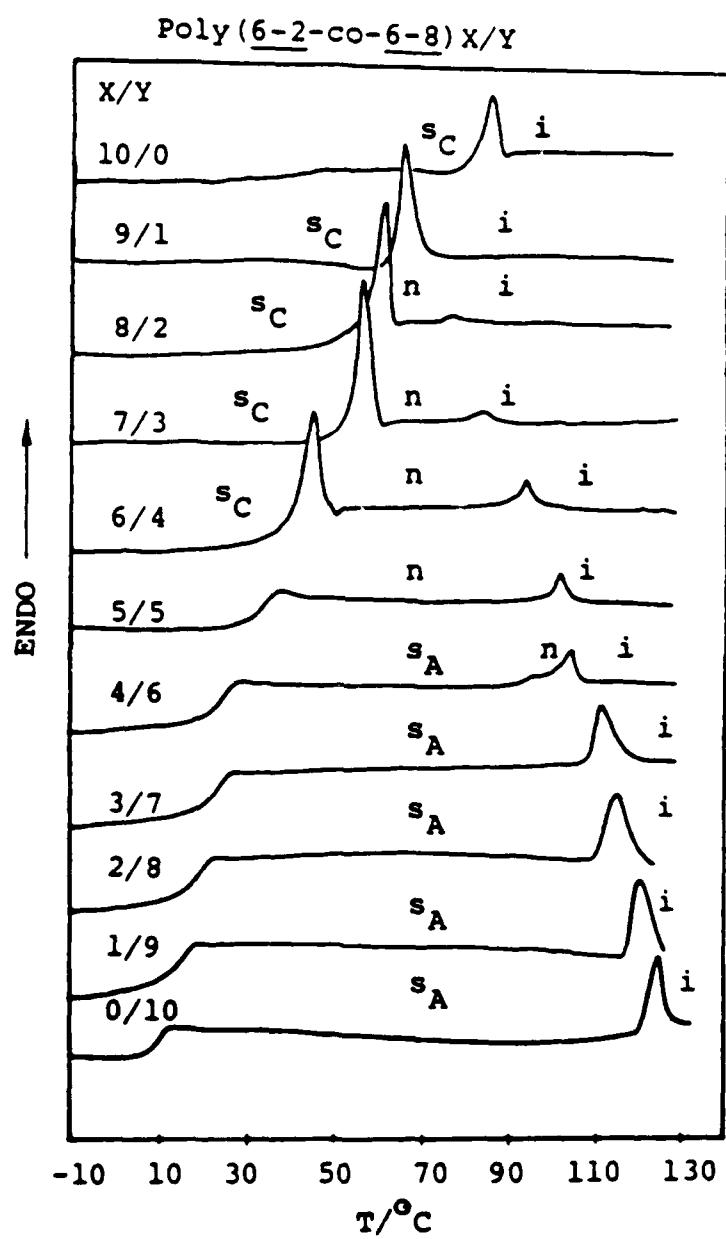


Figure 1a

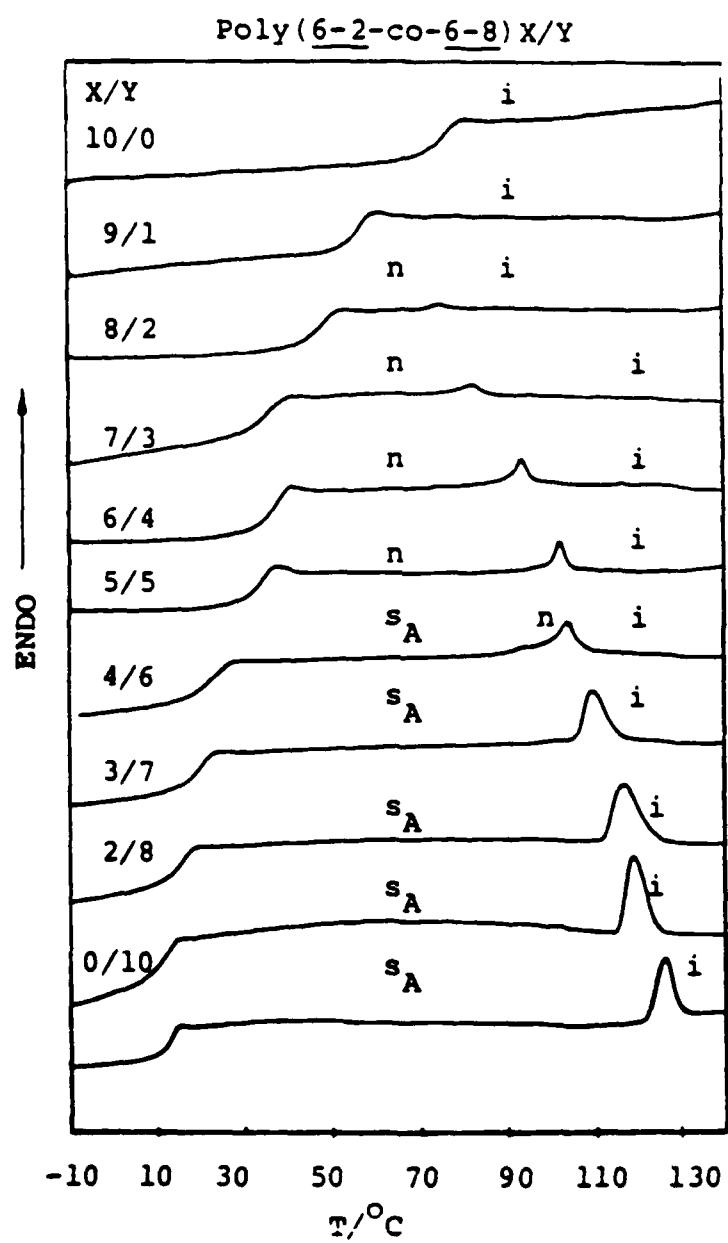


Figure 1b

Poly(6-2-co-6-8) X/Y

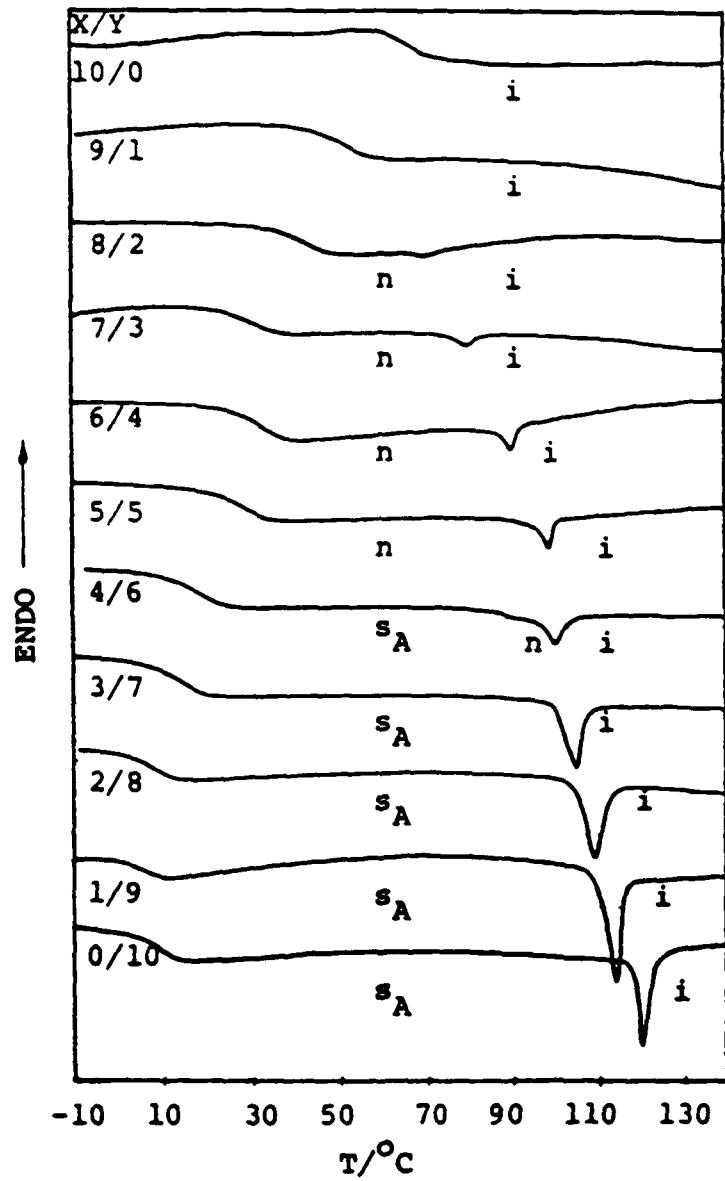


Figure 1c

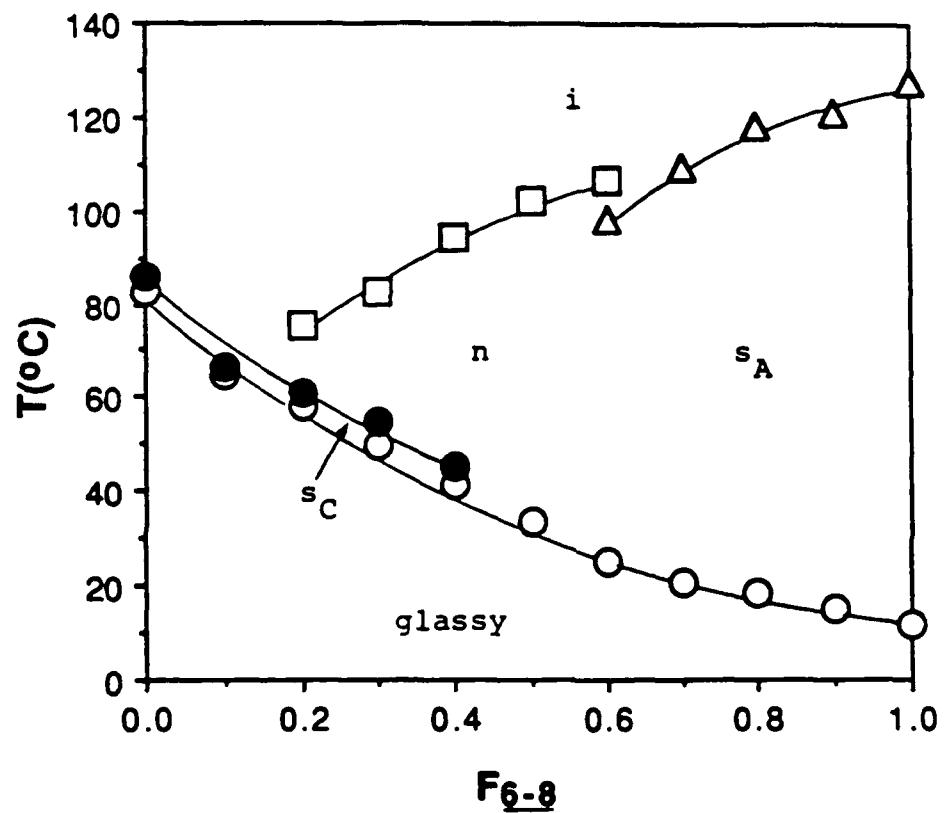


Figure 2a

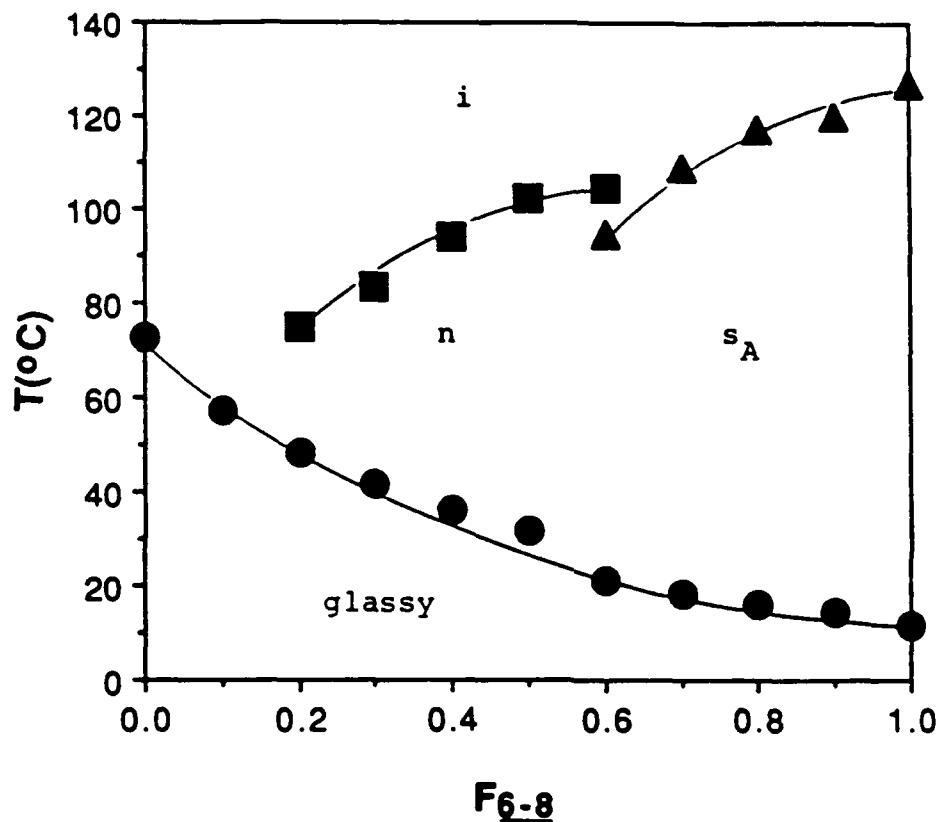


Figure 2b

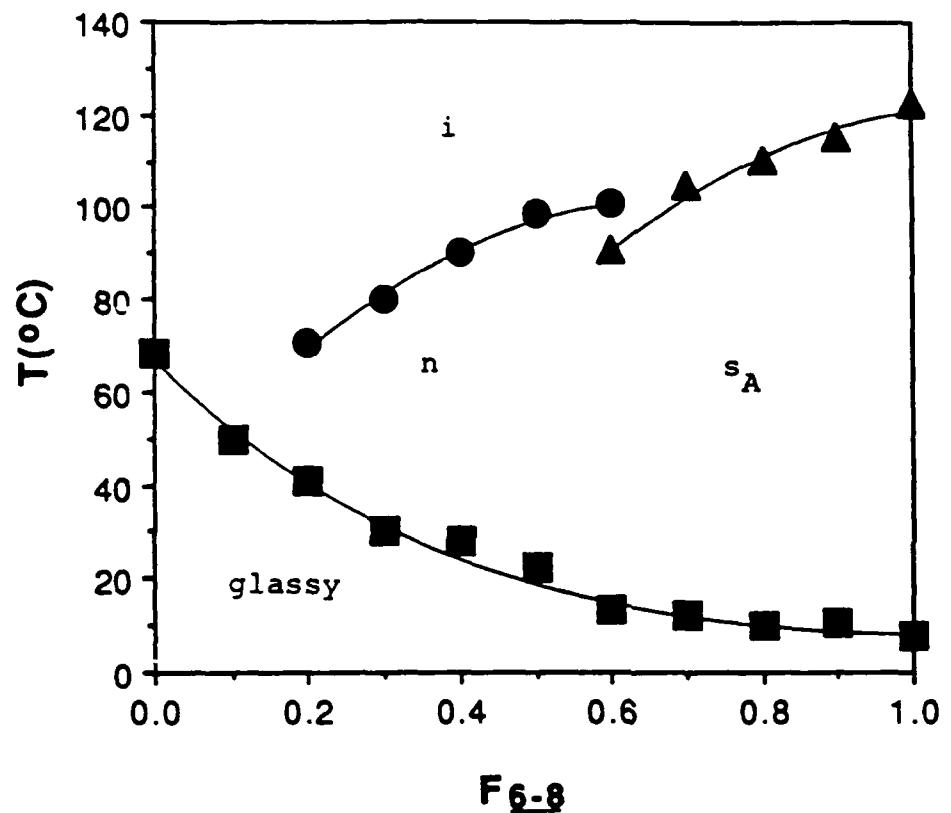


Figure 2c

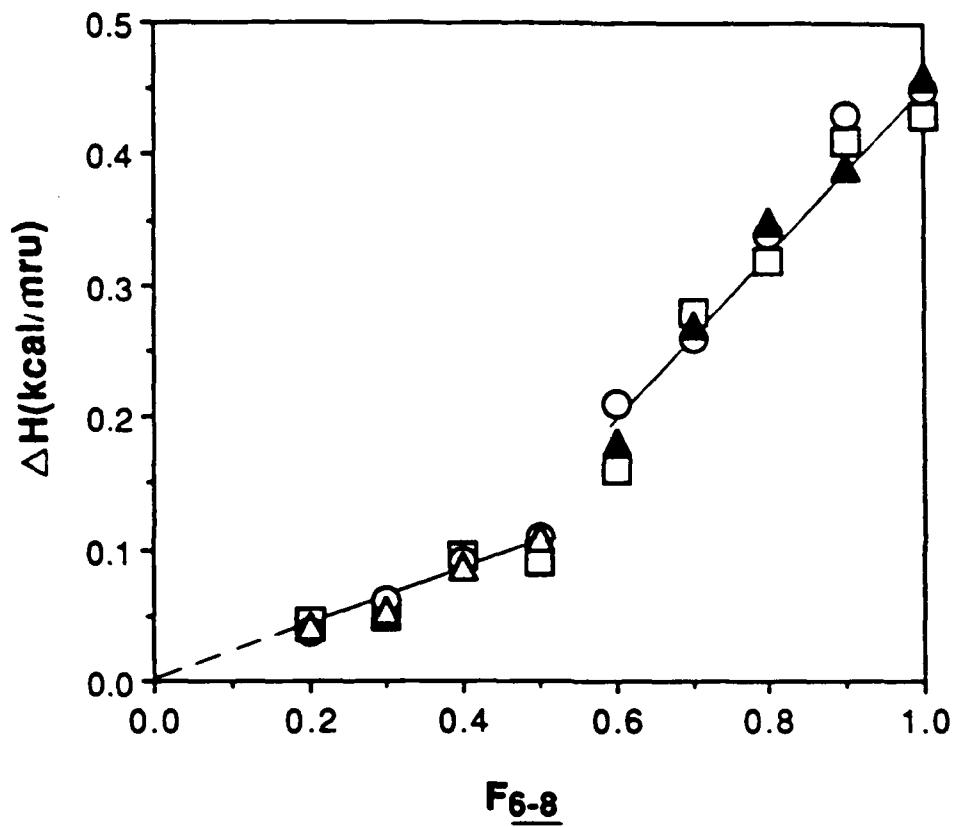


Figure 3

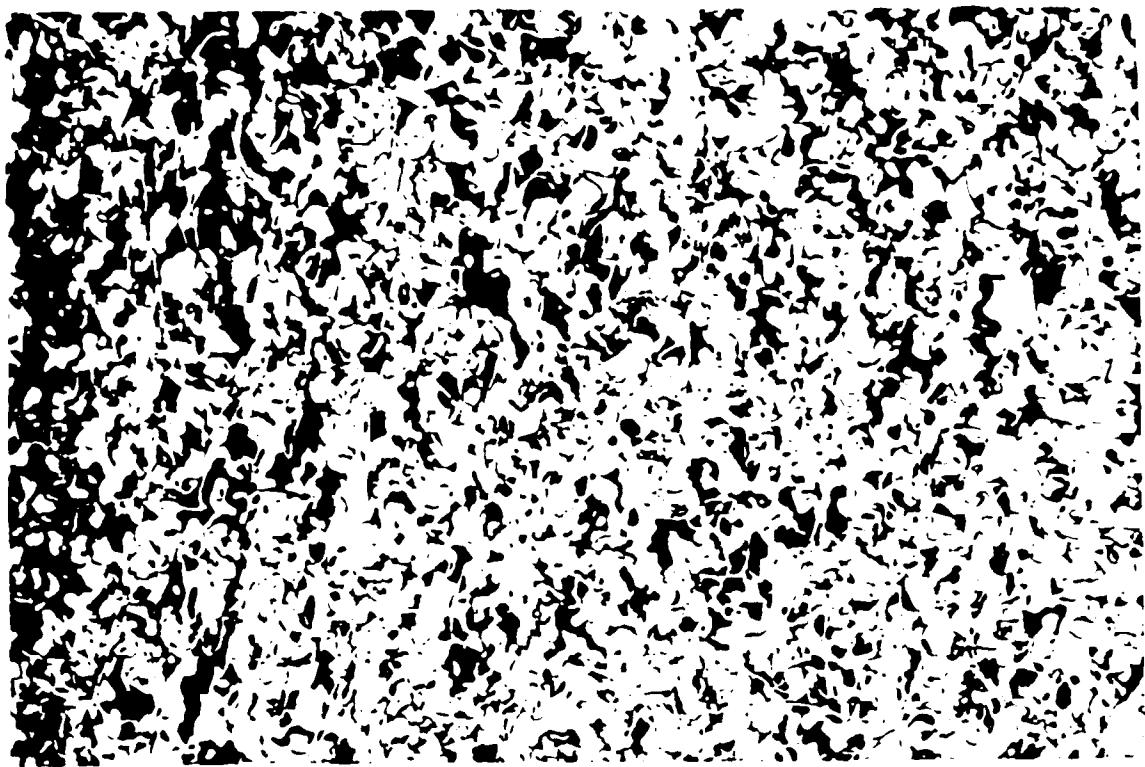


Figure 4a

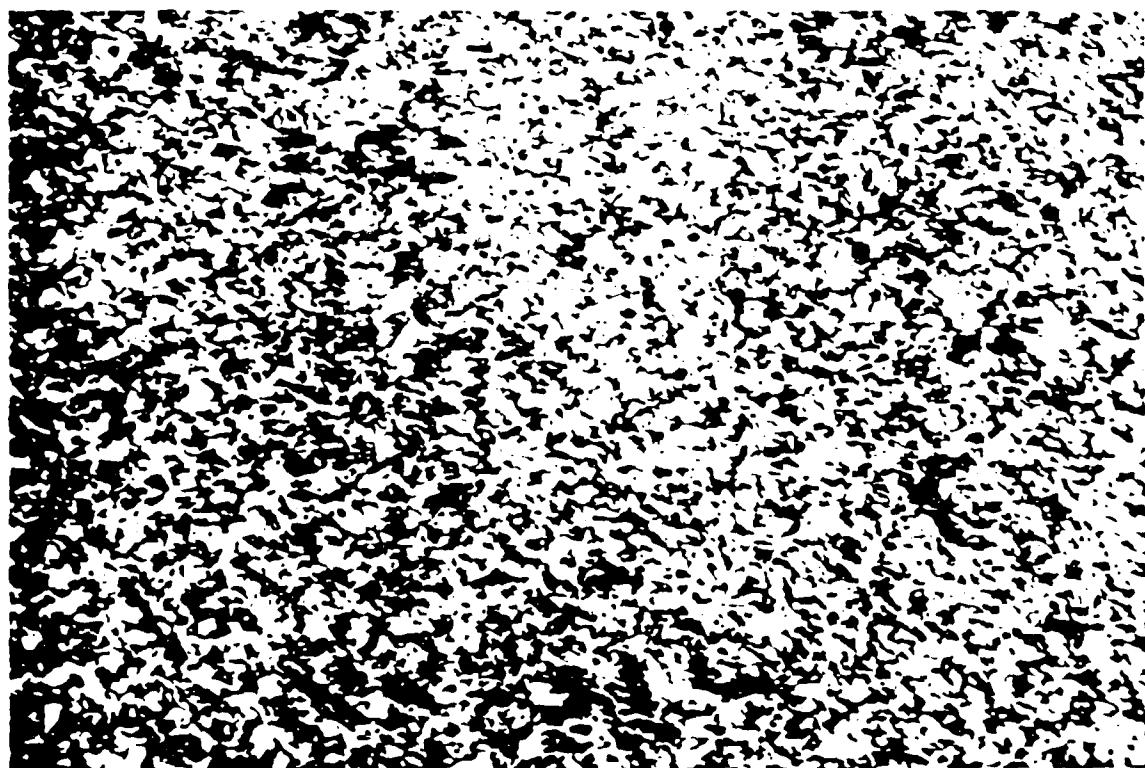


Figure 4b

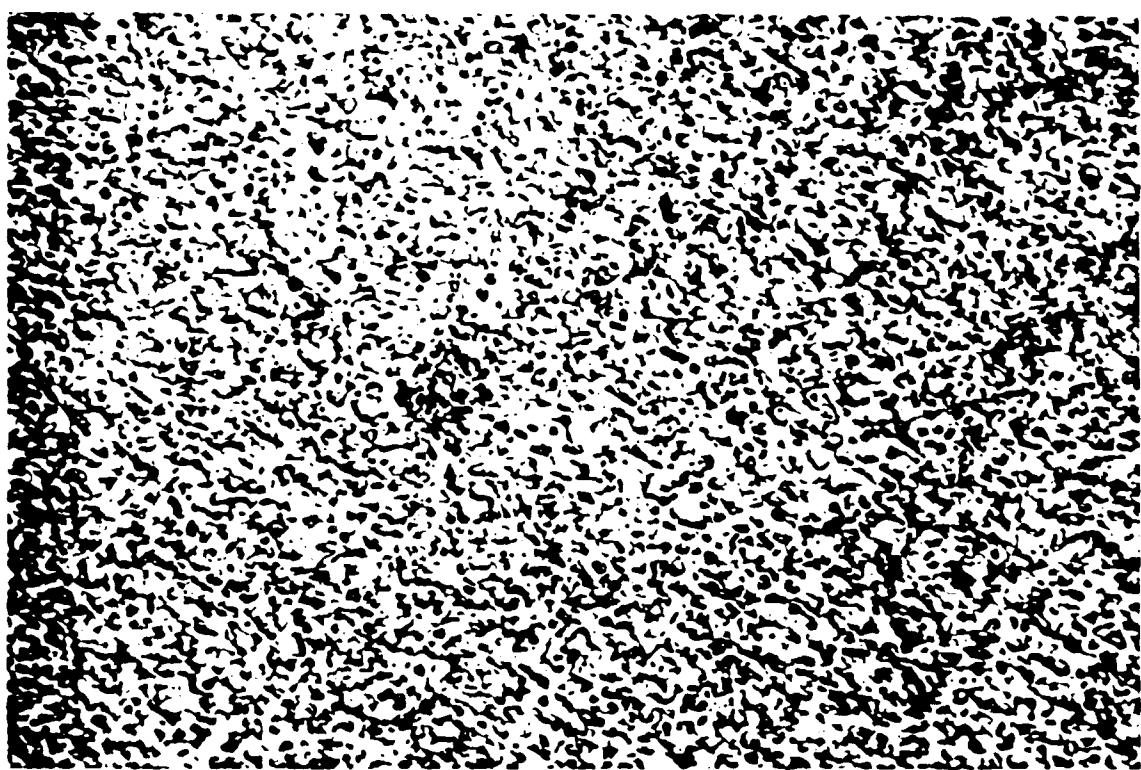


Figure 4c